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**Vinzenz Unger**

Department of molecular Biophysics and Biochemistry

Yale University

P.O. Box 208024

New Haven, CT 06520-8024

Tel: (203) 785-5652

Fax: (203) 785-6404

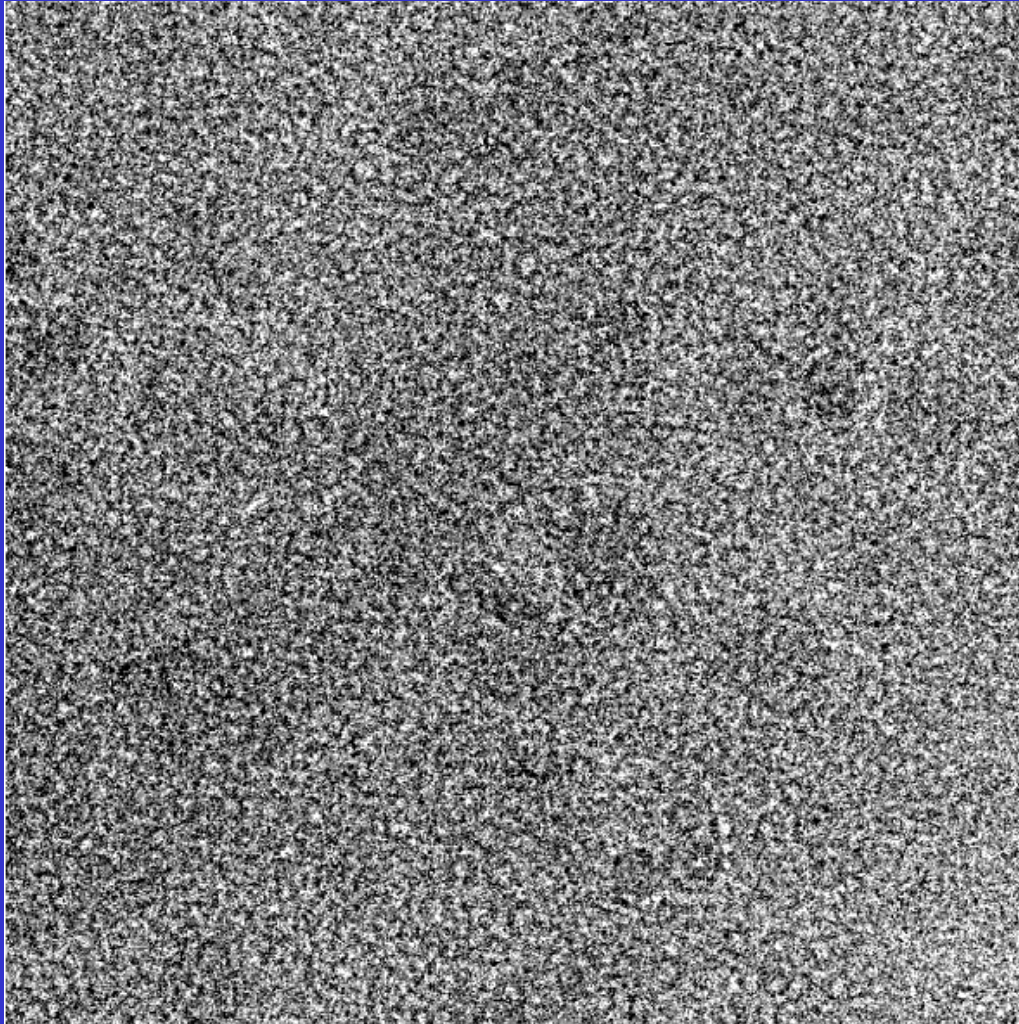
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# Electron Crystallography of Two-Dimensional Crystals

Image Processing - the Basics

V. Unger, 11/16/03

# Low-Dose Image of a Vitrified Gap-Junction 2D-Crystal

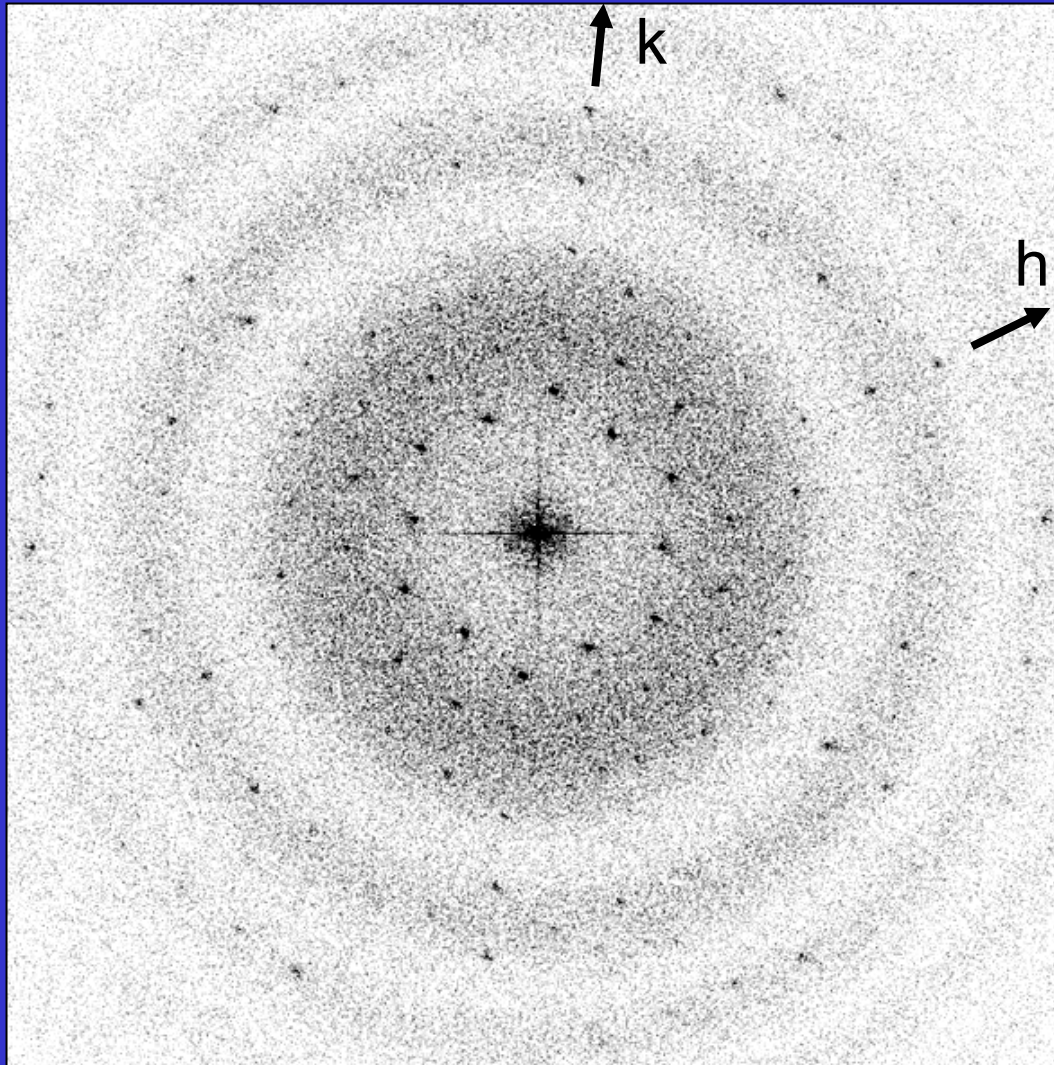


Exposure was  $\sim 10 \text{ e}/\text{\AA}^2$

The low electron dose causes the image to be very noisy.

Therefore, averaging of information from many particles is necessary.

# Calculated Fourier Transform of a Vitrified Gap-Junction 2D-Crystal

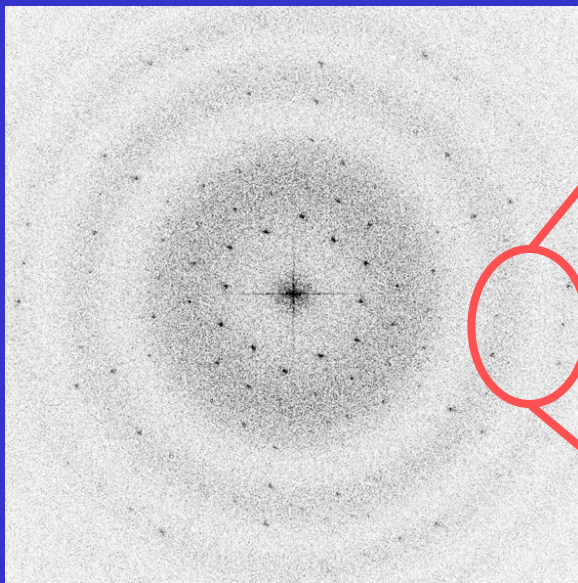


Transform was calculated from image area of 2000x2000 pixel @ 10 $\mu$ m/each

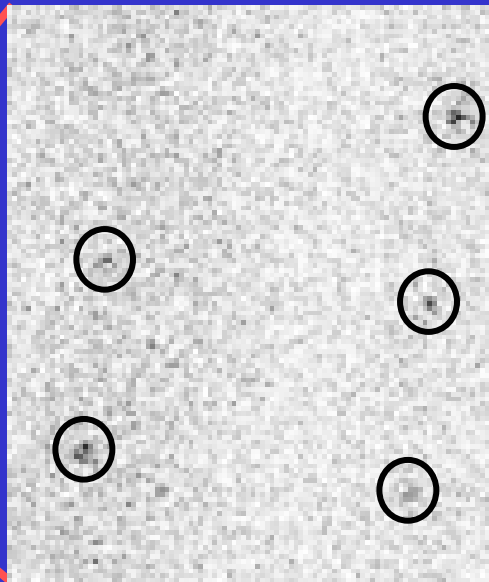
In contrast to X-ray diffraction patterns, diffraction patterns calculated from images contain both amplitude AND phase information for each reflection!

The alternating pattern of bright and dark bands stems from the amorphous carbon film used to support the specimen and reflects a modulation of the diffraction pattern introduced by the objective lens. This modulation is also known as contrast transfer function (CTF) and *per se* has nothing to do with the structure of the molecule whose structure one wants to determine.

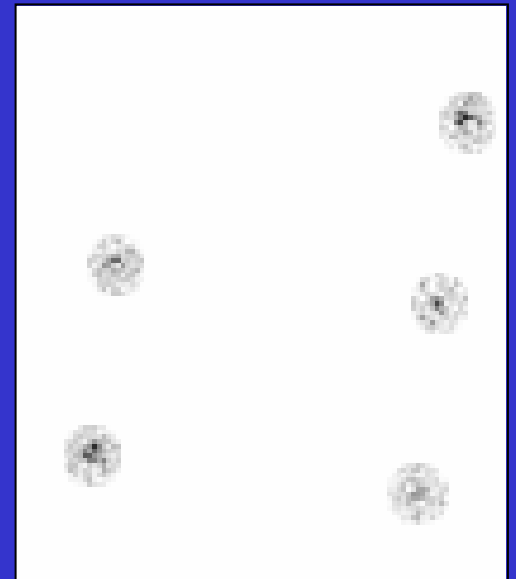
# Principle of Digital Filtering



entire FT



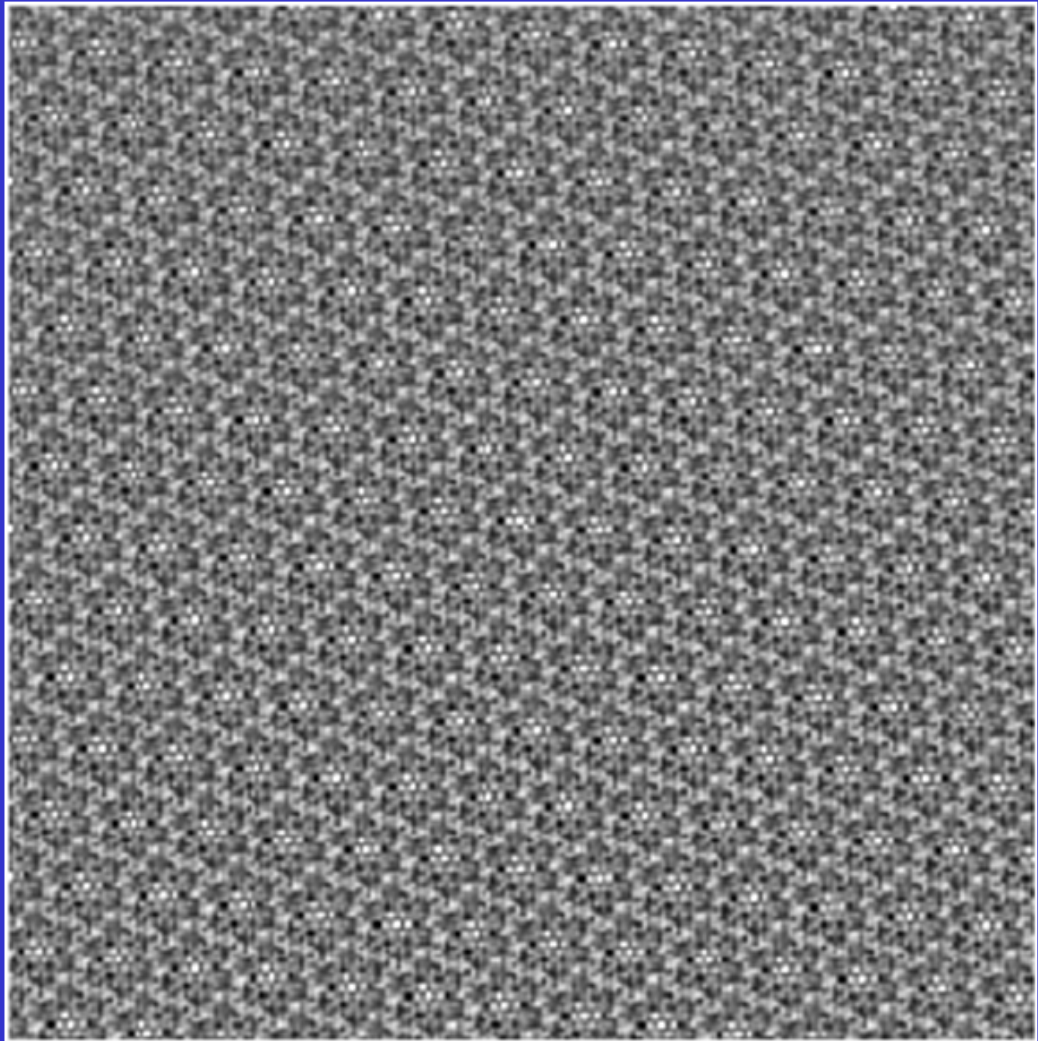
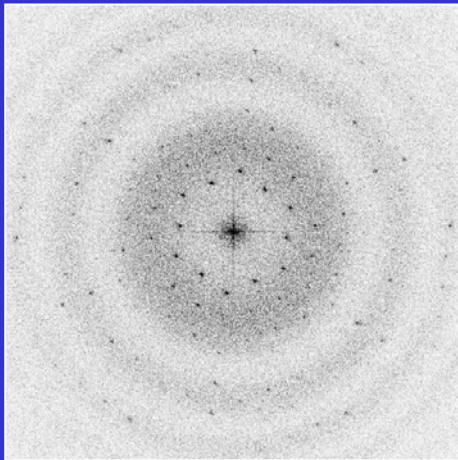
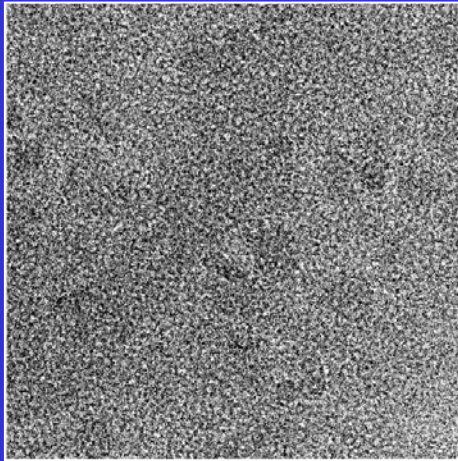
enlarged area of FT



circular maskholes  
applied  
(FT has now non-zero  
values only within  
maskholes)

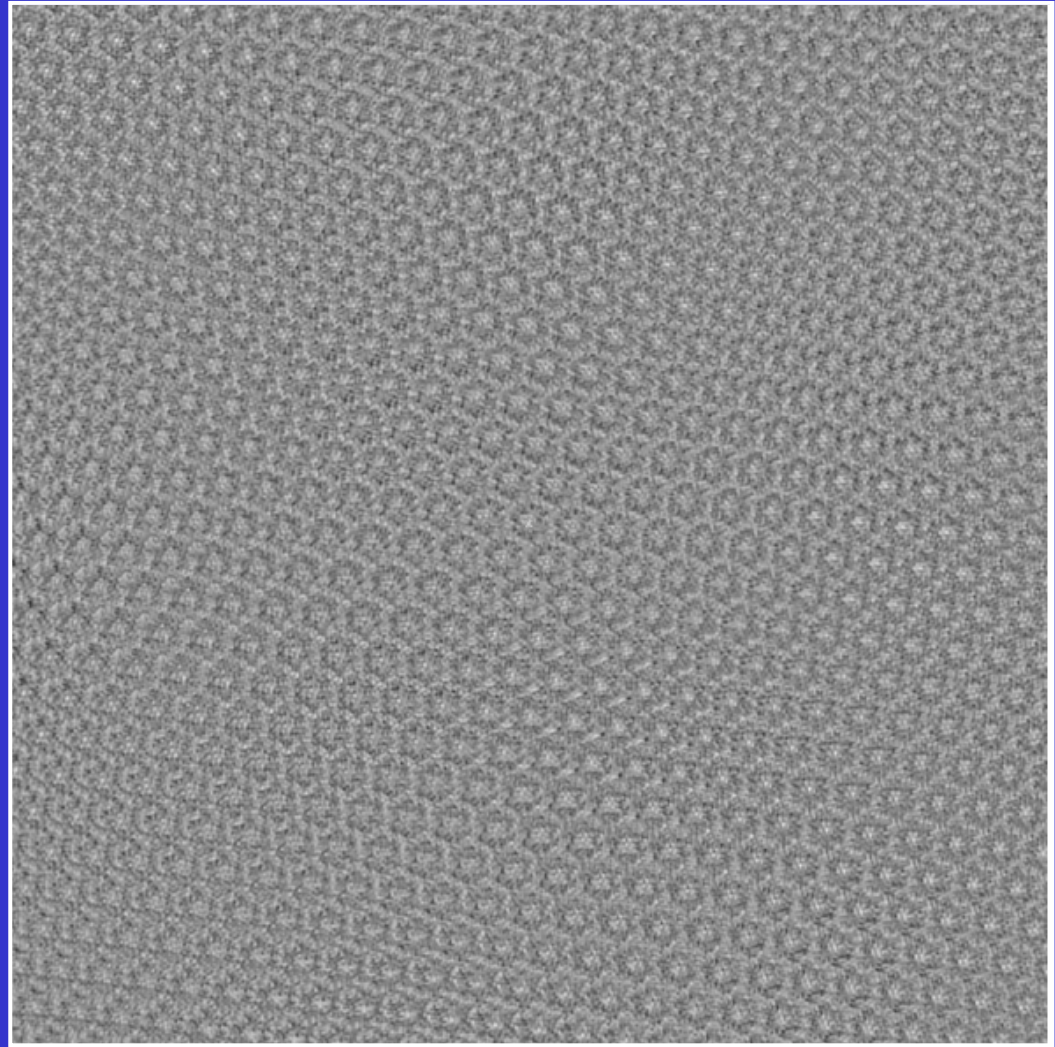
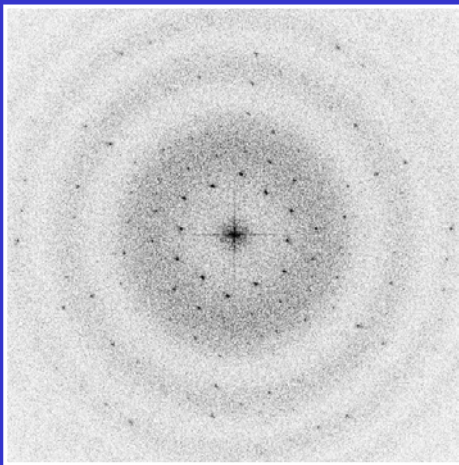
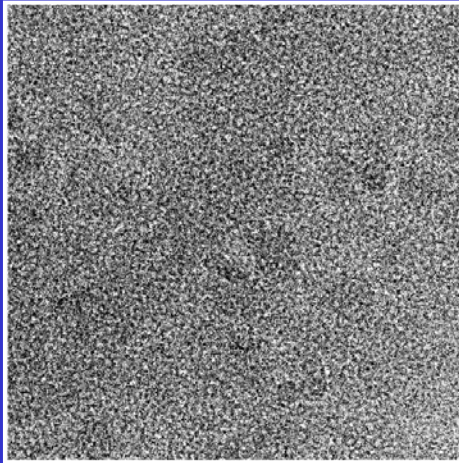
# Digital Filtering of Fourier Transform

Radius used was  $r=1$



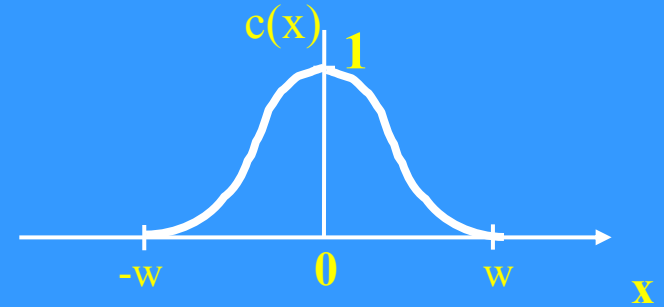
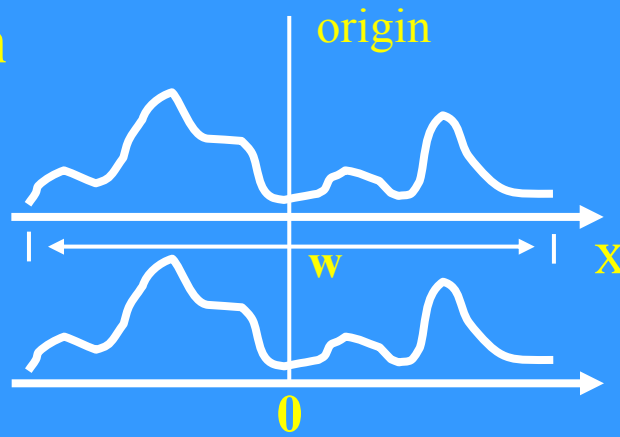
# Digital Filtering of Fourier Transform

Radius used was  $r=7$



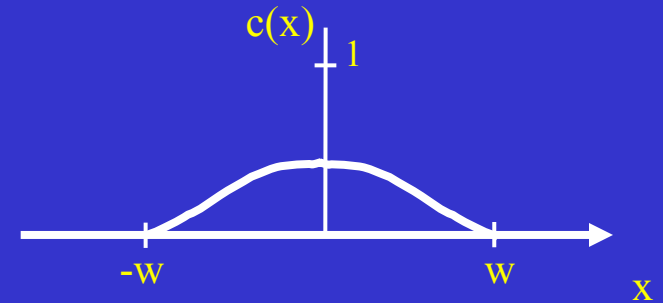
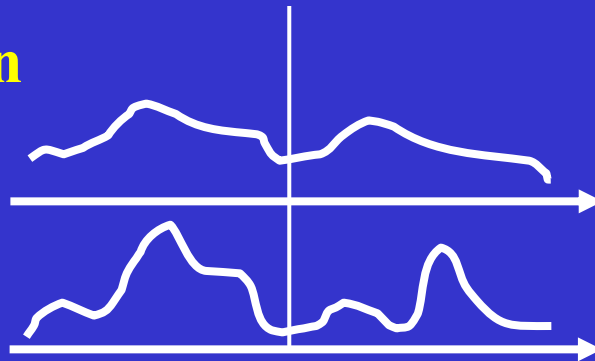
## autocorrelation

two copies of  
same object

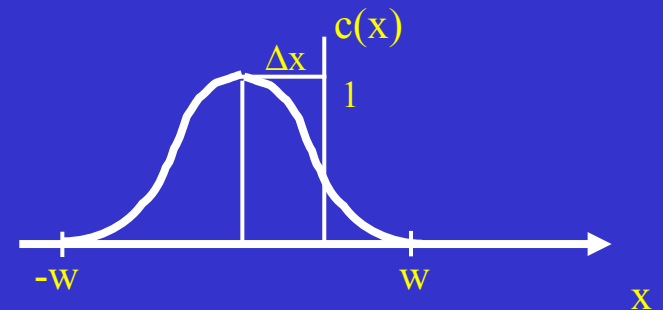
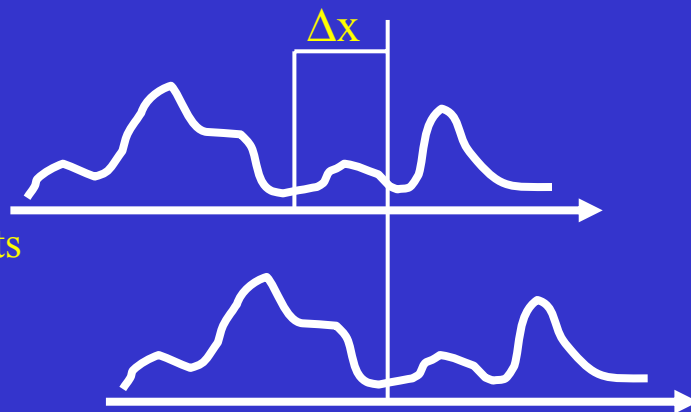


## crosscorrelation

two similar objects

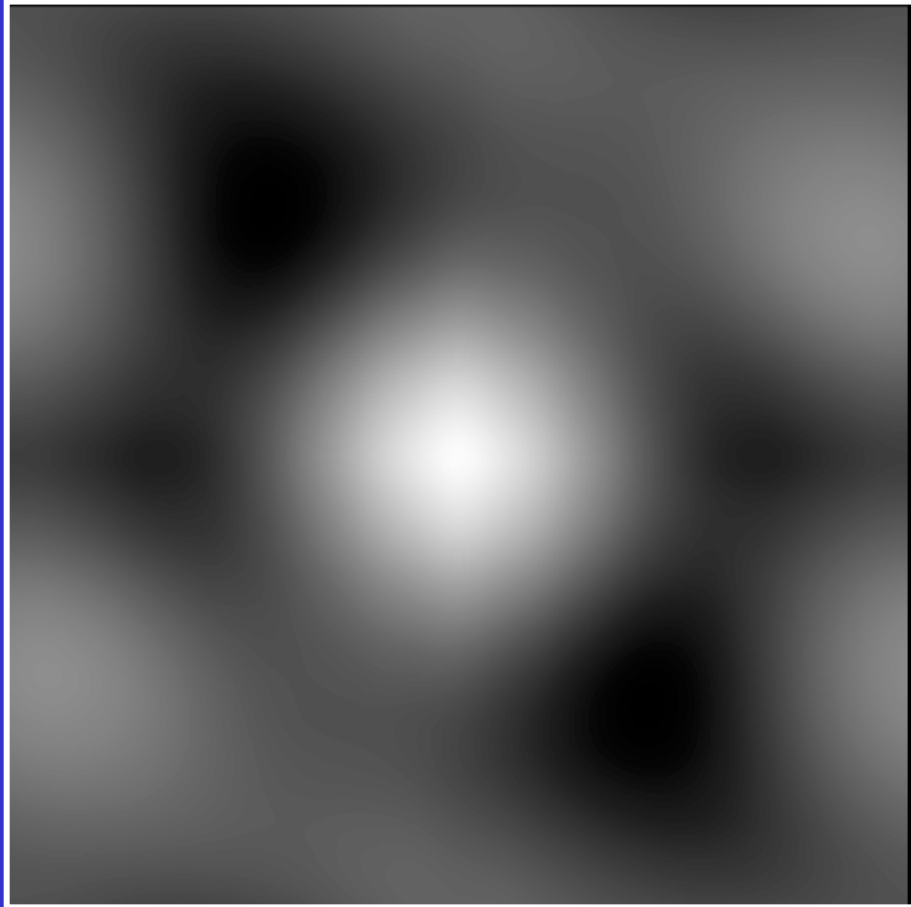


two identical objects  
with translational  
offset

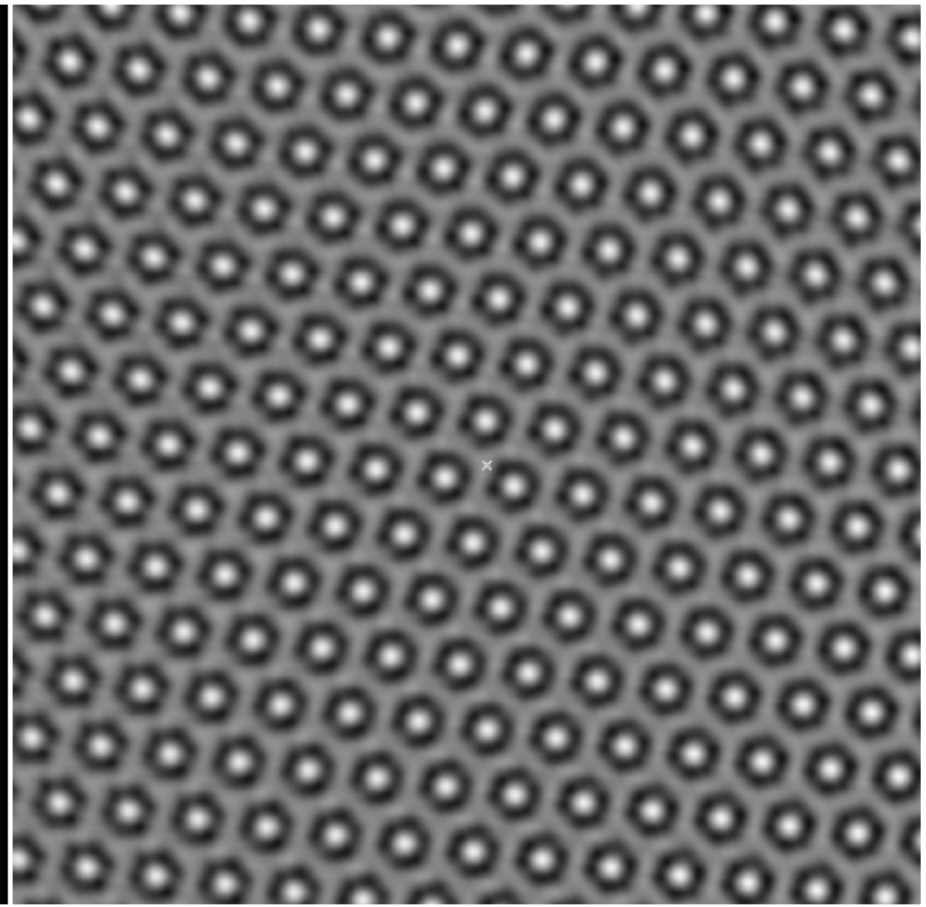




**Autocorrelation Map**



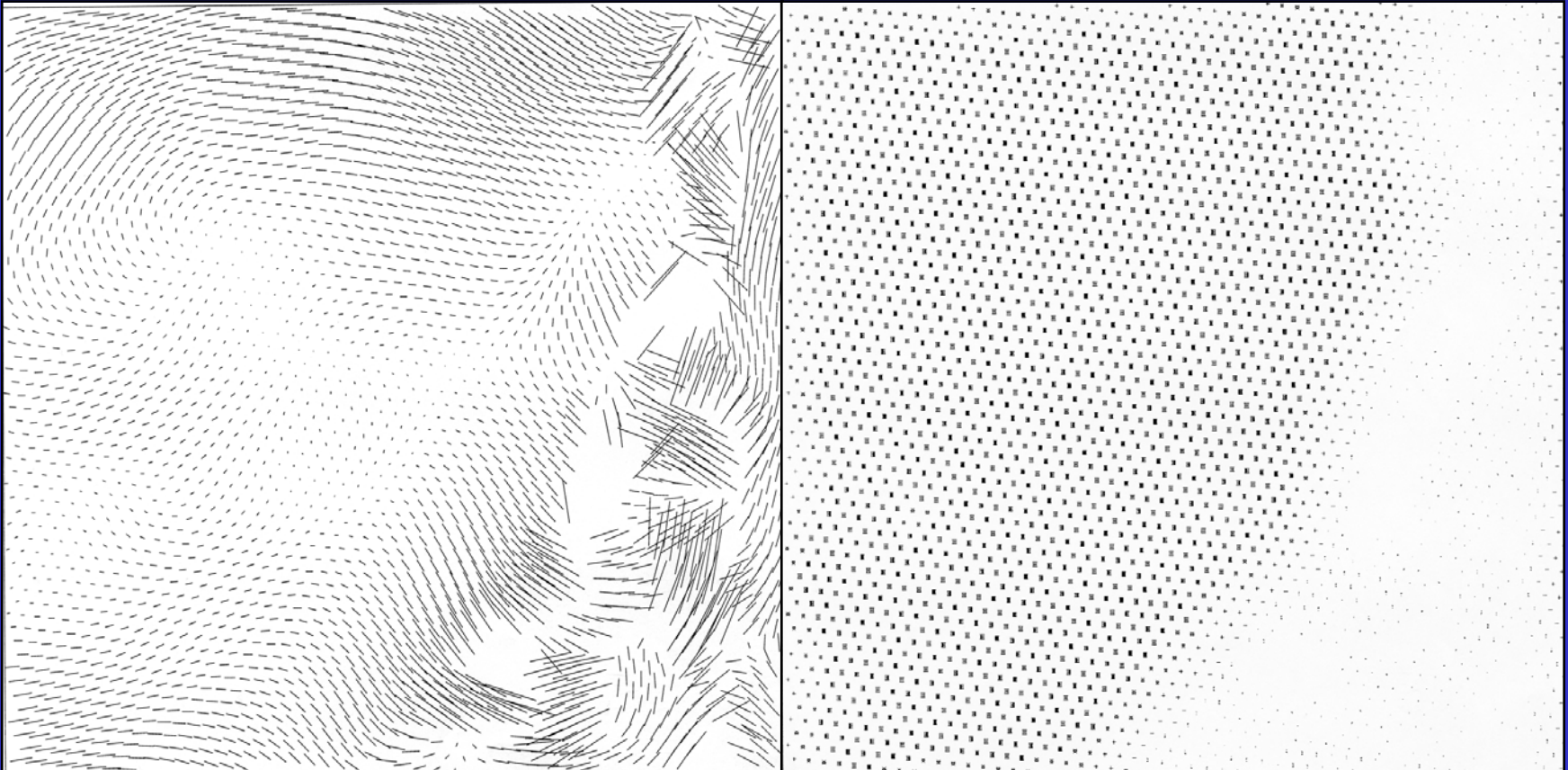
**Part of Crosscorrelation Map**



**Note that the shape of the central peak in the autocorrelation map is very similar to the shape of the cross-correlation peaks.**

# Crosscorrelation Maps

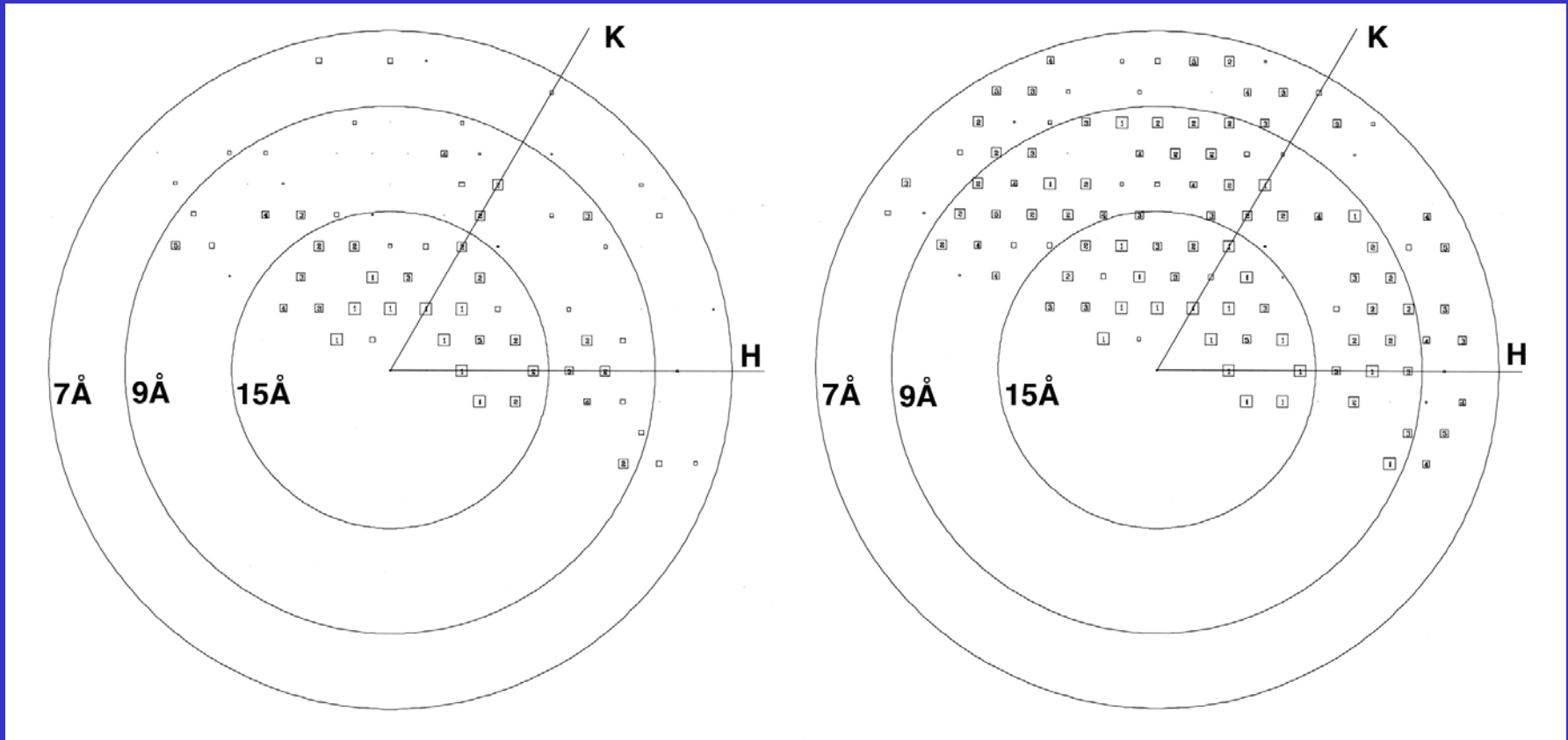
Cross-correlation methods can be used to determine translational disorder in 2D-crystals.



deviation from expected lattice position [ $\text{\AA}$ ] X20 (not to scale) with respect to chosen reference

height of cross correlation-peaks indicates how similar each unit cell is to the chosen reference

## Effect of “Lattice Unbending”

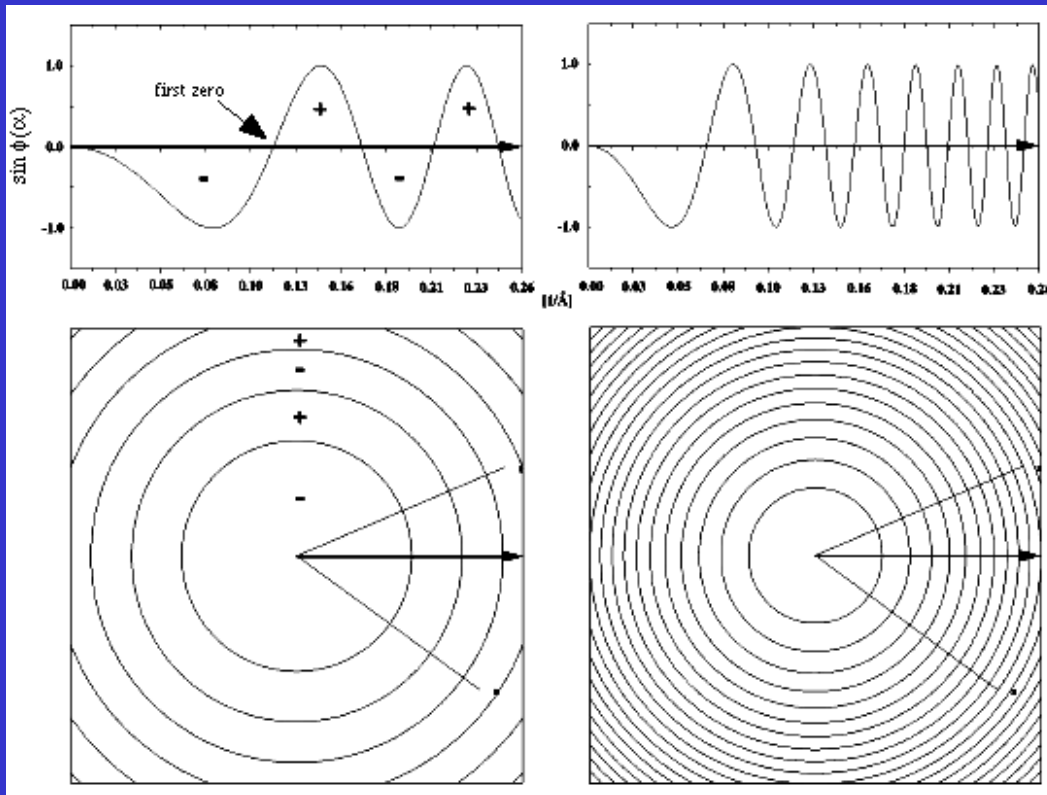


The data shown to the left were retrieved from a calculated FT of an untreated raw image. In this case, the data are not statistically significant beyond  $\sim 15\text{\AA}$  resolution. However, after correction for translational lattice disorder, the same image provides data out to  $\sim 7\text{\AA}$  resolution. Plot symbols indicate the goodness of each reflection. Reflections marked by a “1” have a signal-to-noise ratio of at least 8.

The phase shift  $\Phi(\alpha)$  introduced to the scattered waves depends on the wavelength ( $\lambda$ ), spherical aberration of the objective lens  $C_s$ , the diffraction angle ( $\alpha$ ) and the underfocus ( $\Delta f$ ).

$$\Phi(\alpha) = \frac{2\pi}{\lambda} (0.25 \cdot C_s \alpha^4 - 0.5 \cdot \Delta f \alpha^2)$$

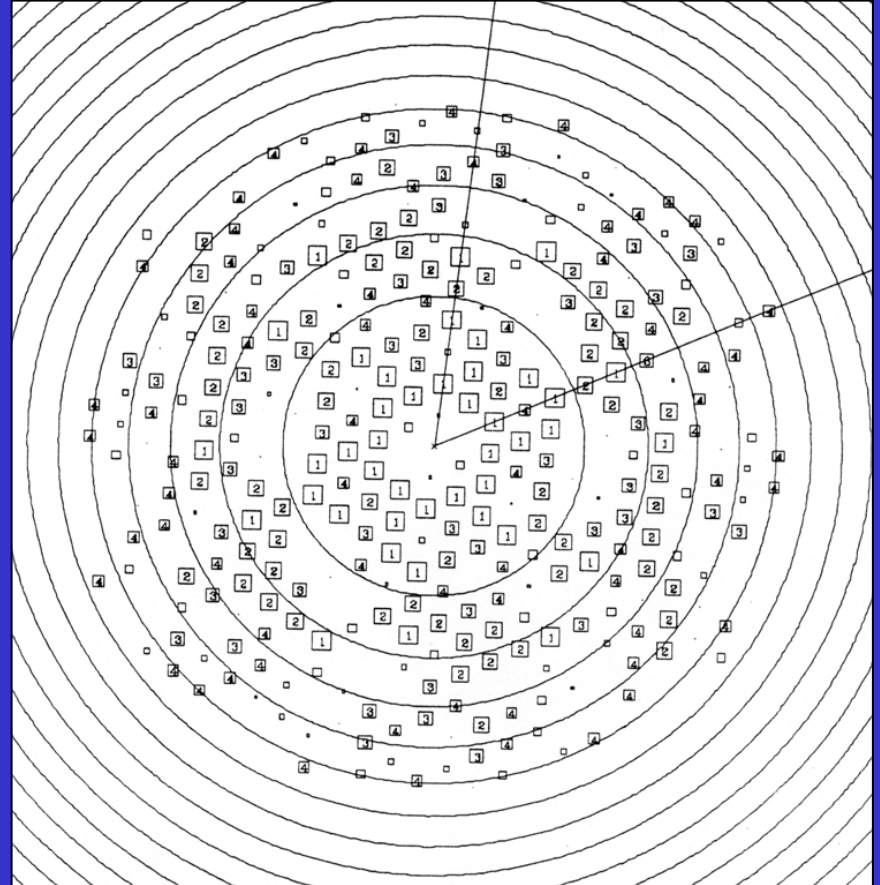
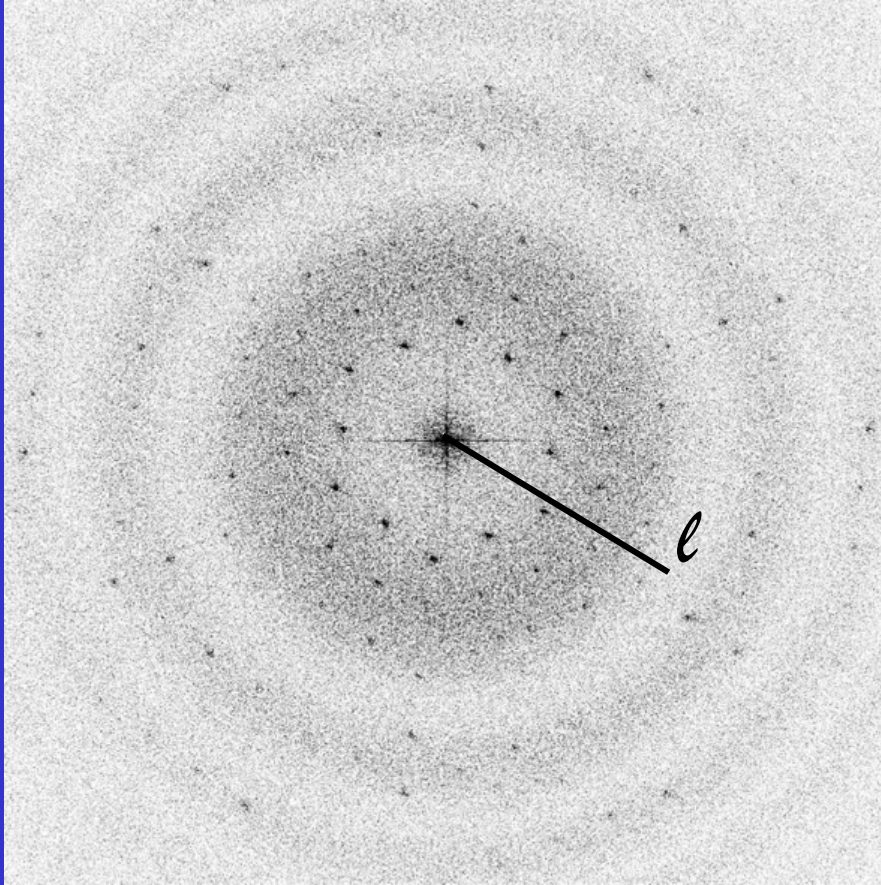
This phase shift (also known as CTF= contrast transfer function) modulates the FT of the image in an oscillating manner described by  $[\sin \Phi(\alpha)]$ .



The simulated curves are for 3000 and 6000Å of underfocus respectively, an accelerating voltage of 200keV ( $\lambda=0.025\text{\AA}$ ) and a  $C_s=2\text{mm}$

These lower two panels demonstrate how the CTF would look like in the FT of the image. Circles represent  $[\sin \Phi(\alpha)] = 0$  Frequencies where  $[\sin \Phi(\alpha)] < 0$  contribute with reversed contrast to the image. Therefore, the phases of reflections in these regions need to be adjusted by  $180^\circ$

# CTF-correction



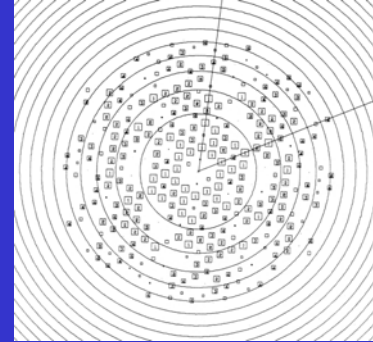
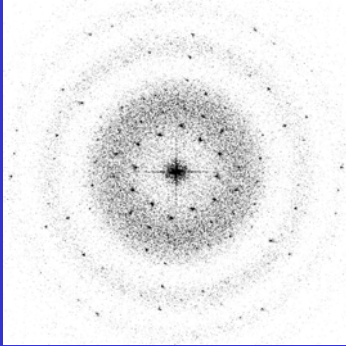
## Calculated FT of image

$$1/d[\text{\AA}^{-1}] = \frac{\ell \cdot XMAG}{p_y \cdot step[\mu m] \cdot 10,000}$$

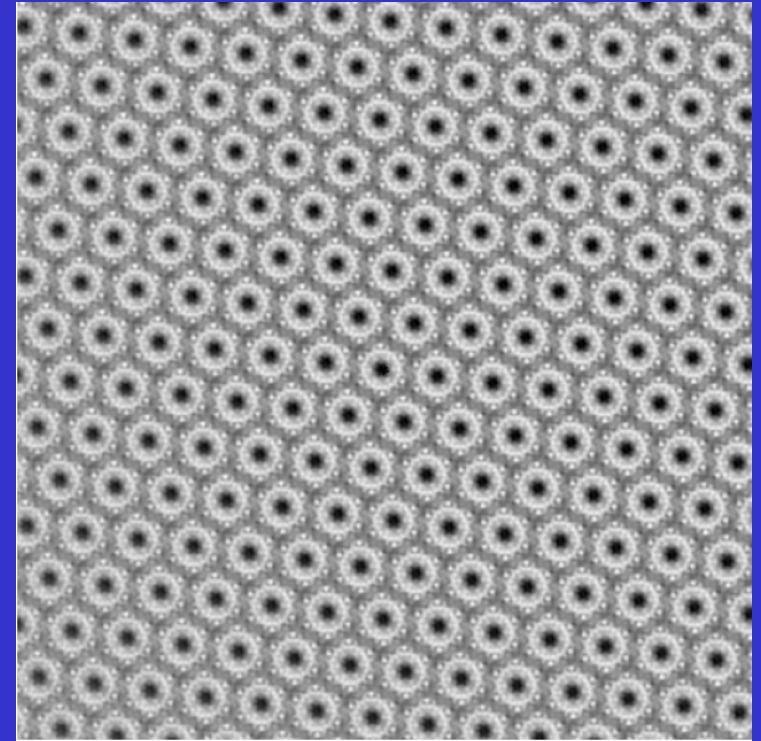
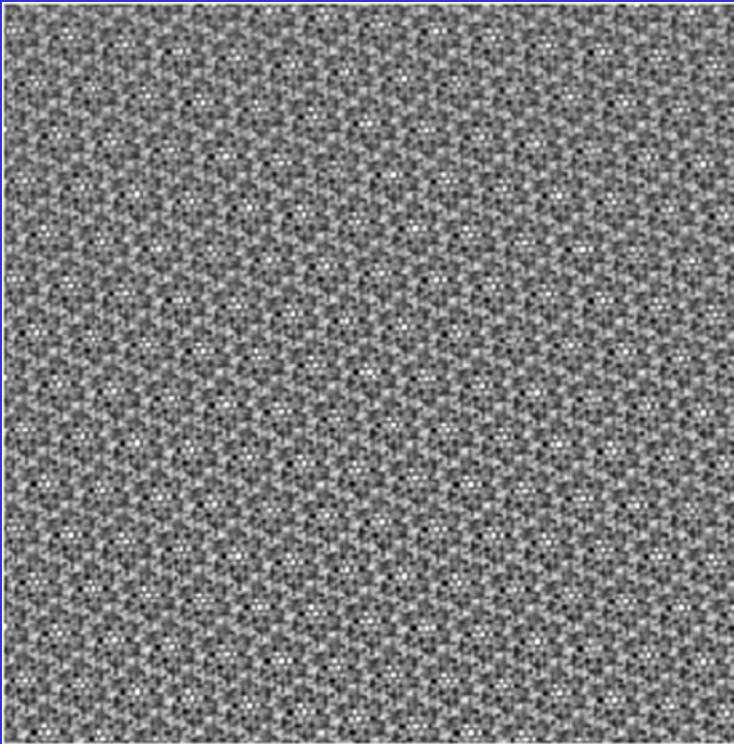
Xmag: magnification  
 $p_y$ : transform size (y-axis)  
step: digitizing stepsize

For the correct estimate of the underfocus, the simulated CTF will match that actually observed in the FT of the image

# Why Do Data Need to be Corrected For CTF?

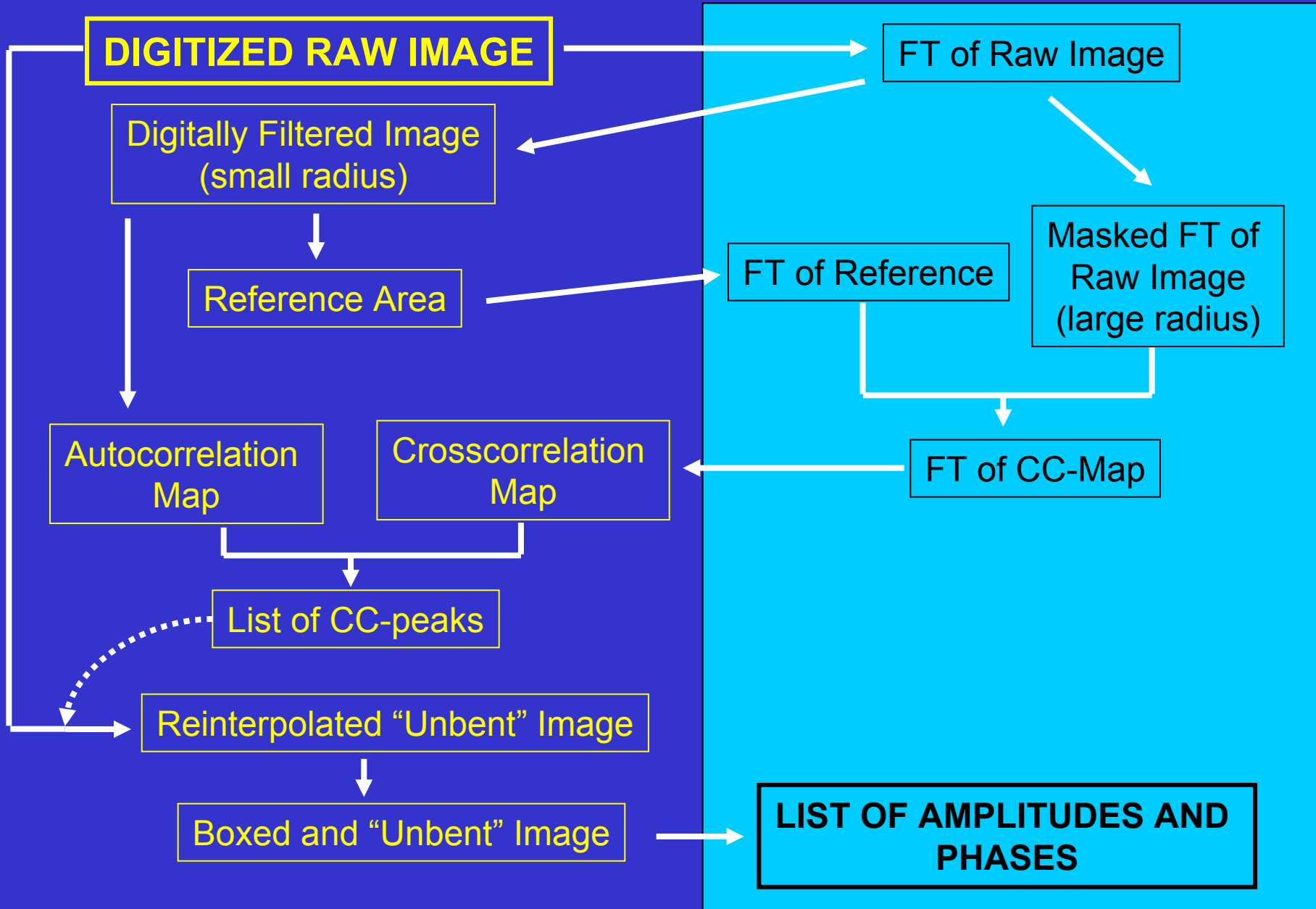


After  
Correction  
of Phases  
For CTF

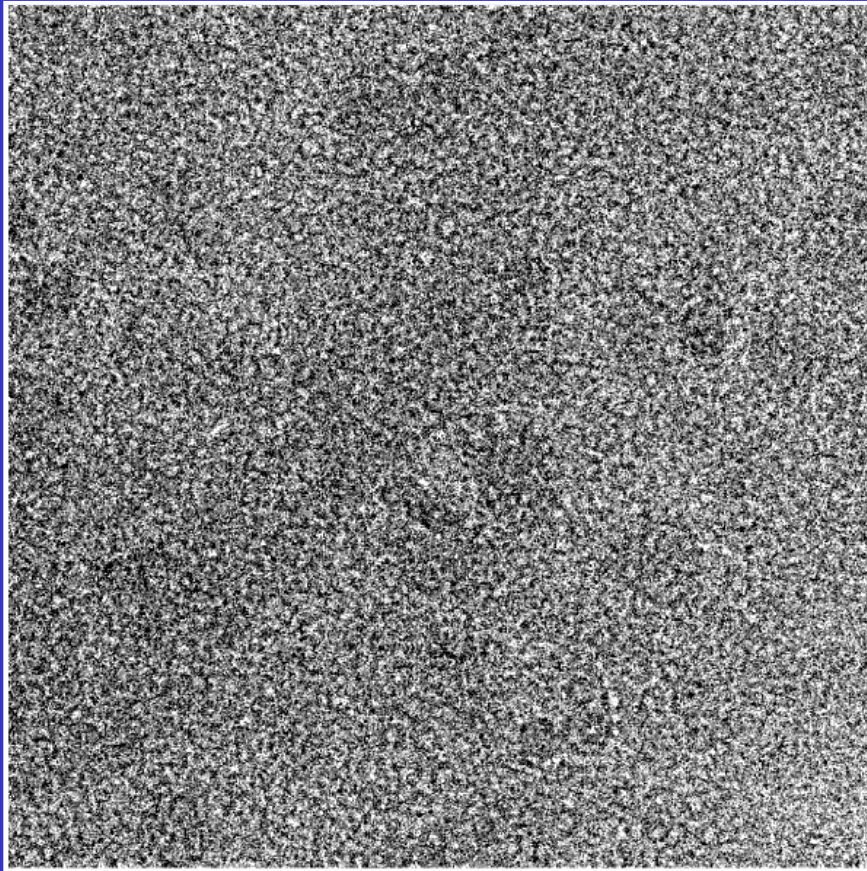


# REAL SPACE

# RECIPROCAL SPACE



# Basic Image Processing of 2D-Crystals



REAL SPACE

(H,K)	amp	phase	IQ	CTF
0 1	132.4	237.8	7	-0.142
0 2	5686.9	299.8	1	-0.540
0 3	195	249.1	6	-0.958
0 4	1067.4	246.1	1	-0.762
0 5	431.0	102.5	2	0.397
0 6	1016.5	356.5	1	0.925
0 7	120.5	243.0	6	-0.602
0 8	0.0	270.7	9	-0.388
0 9	145.5	319.4	4	0.923
0 10	67.2	290.6	6	-0.993
0 11	0.0	270.7	9	0.928
1 0	97.7	132.8	8	-0.148
1 1	7227.8	140.0	1	-0.423
1 2	2582.2	17.1	1	-0.846
1 3	1460.3	266.5	1	-0.957

And so forth....

RECIPROCAL SPACE



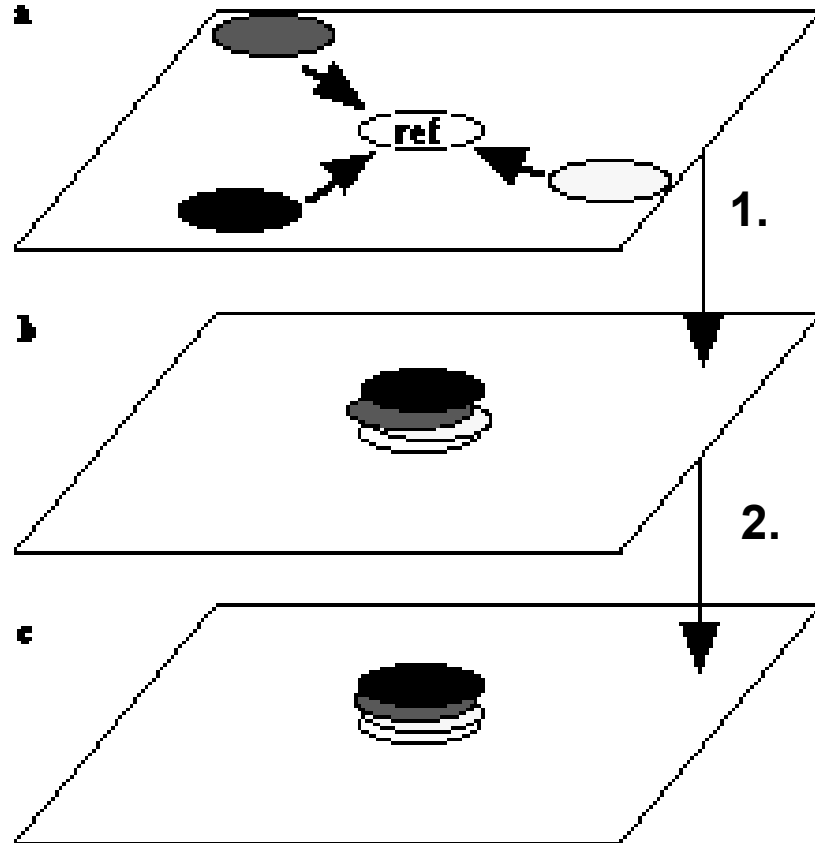


# Symmetry is Beautiful

phase information obtained from the calculated FT of the images contains information about the positioning of molecules within the area of the unit cell as well as information about symmetry.

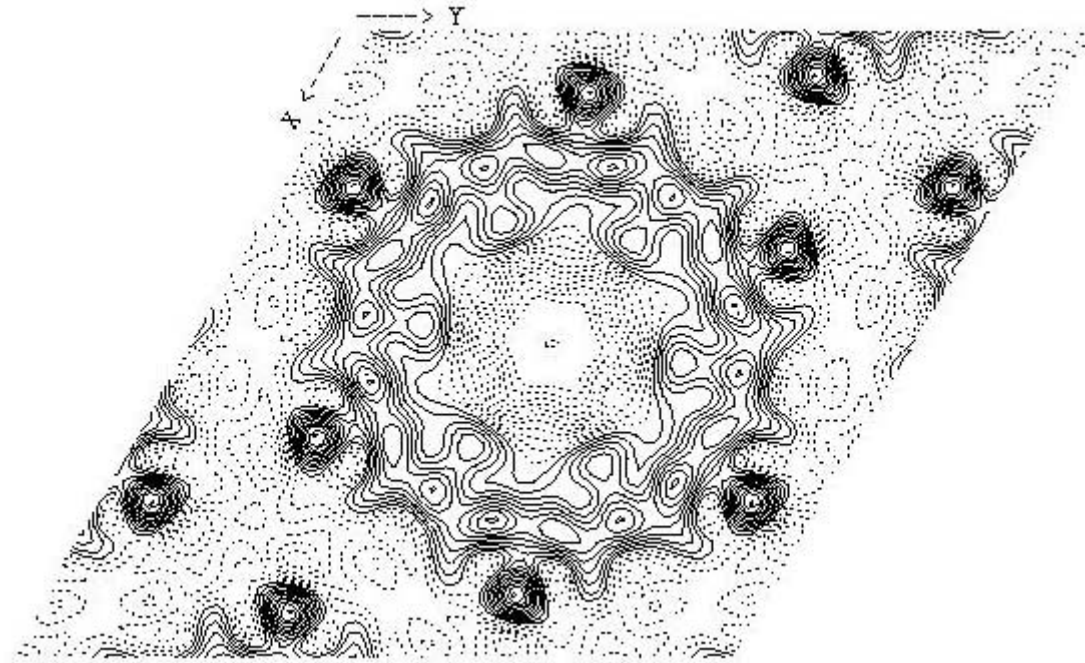
Moreover, presence of crystallographic symmetry facilitates data collection and merging because it causes redundancy in the information. The “mirror symmetry” in faces commonly considered “pretty” is just one odd example.

# Alignment of Data from Independent Images



1. Find common phase origin by comparison with data from single untilted crystal
2. Iteratively merge and refine phase origins of each image against average of all others

# Projection Density Map and some of the Corresponding Structure Factors

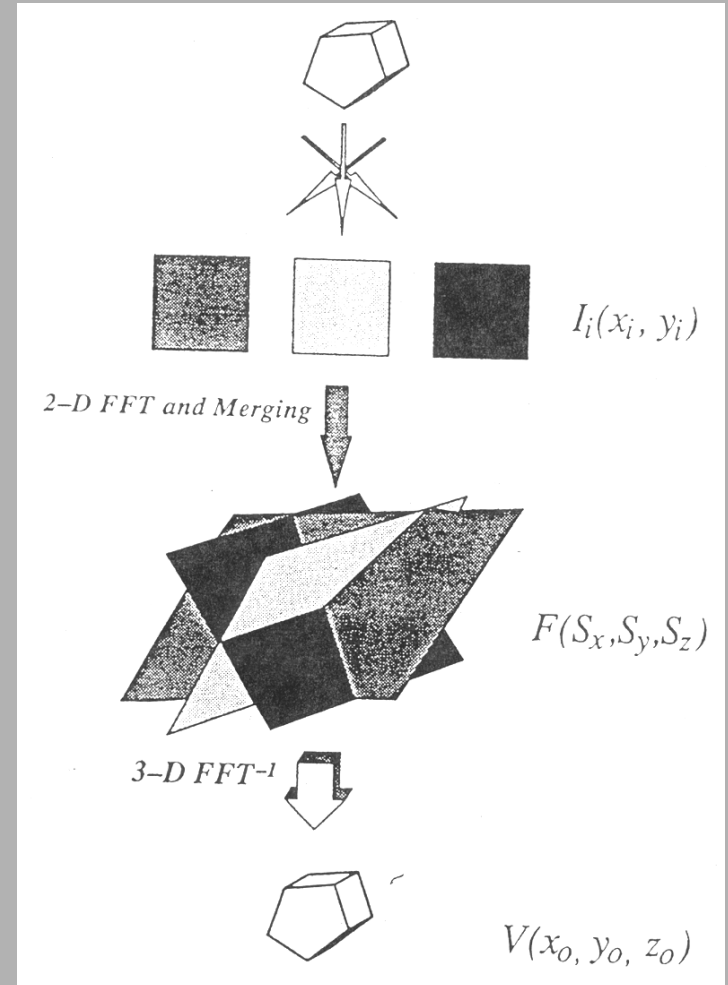
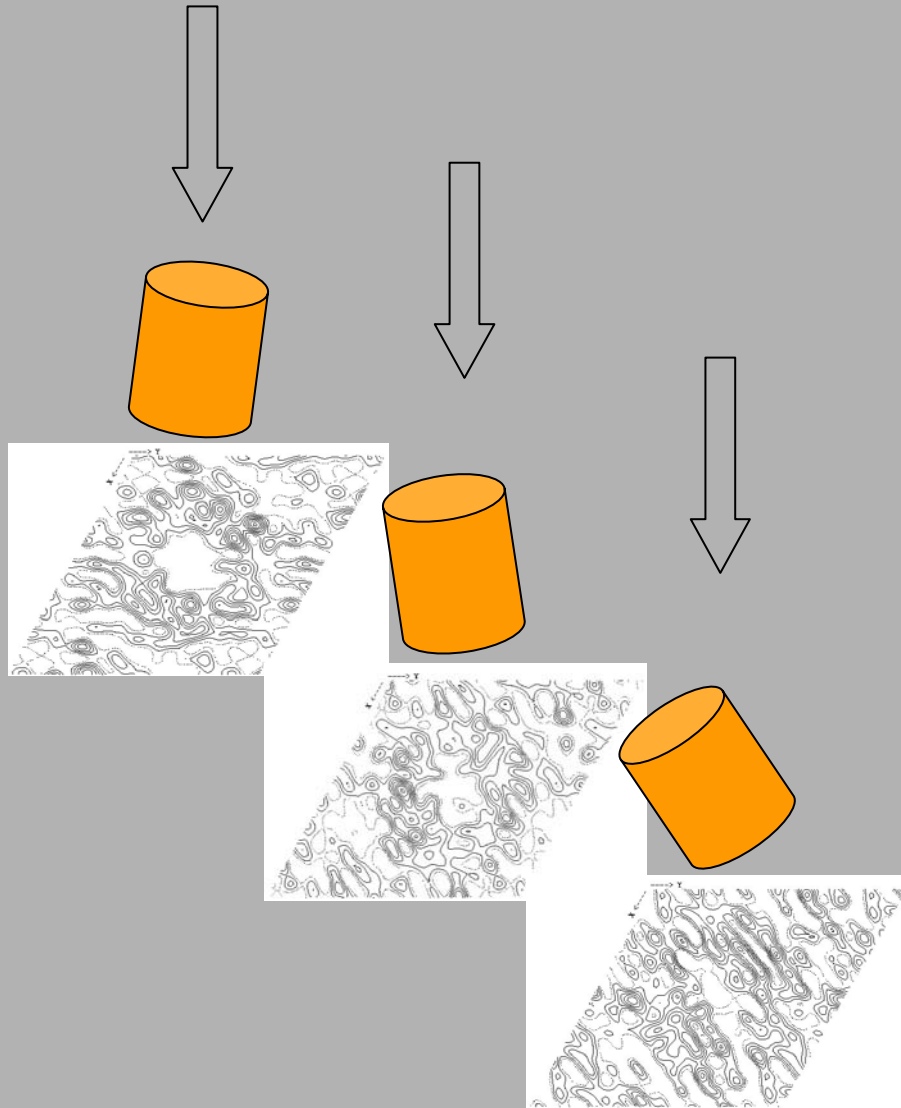


<b>(H,K,L)</b>	<b>amp</b>	<b>phase</b>	<b>FOM</b>
1 0 0	2566	180	99.5
1 1 0	12424	180	99.9
1 2 0	777	180	99.5
1 3 0	1123	0	99.7
1 4 0	208	0	73.9
1 5 0	605	0	99.0
1 6 0	670	180	99.2
1 7 0	250	180	99.6
1 8 0	350	0	94.3
1 9 0	77	180	59.8
1 10 0	140	0	13.3
2 0 0	9265	180	99.9
2 1 0	1971	0	99.8

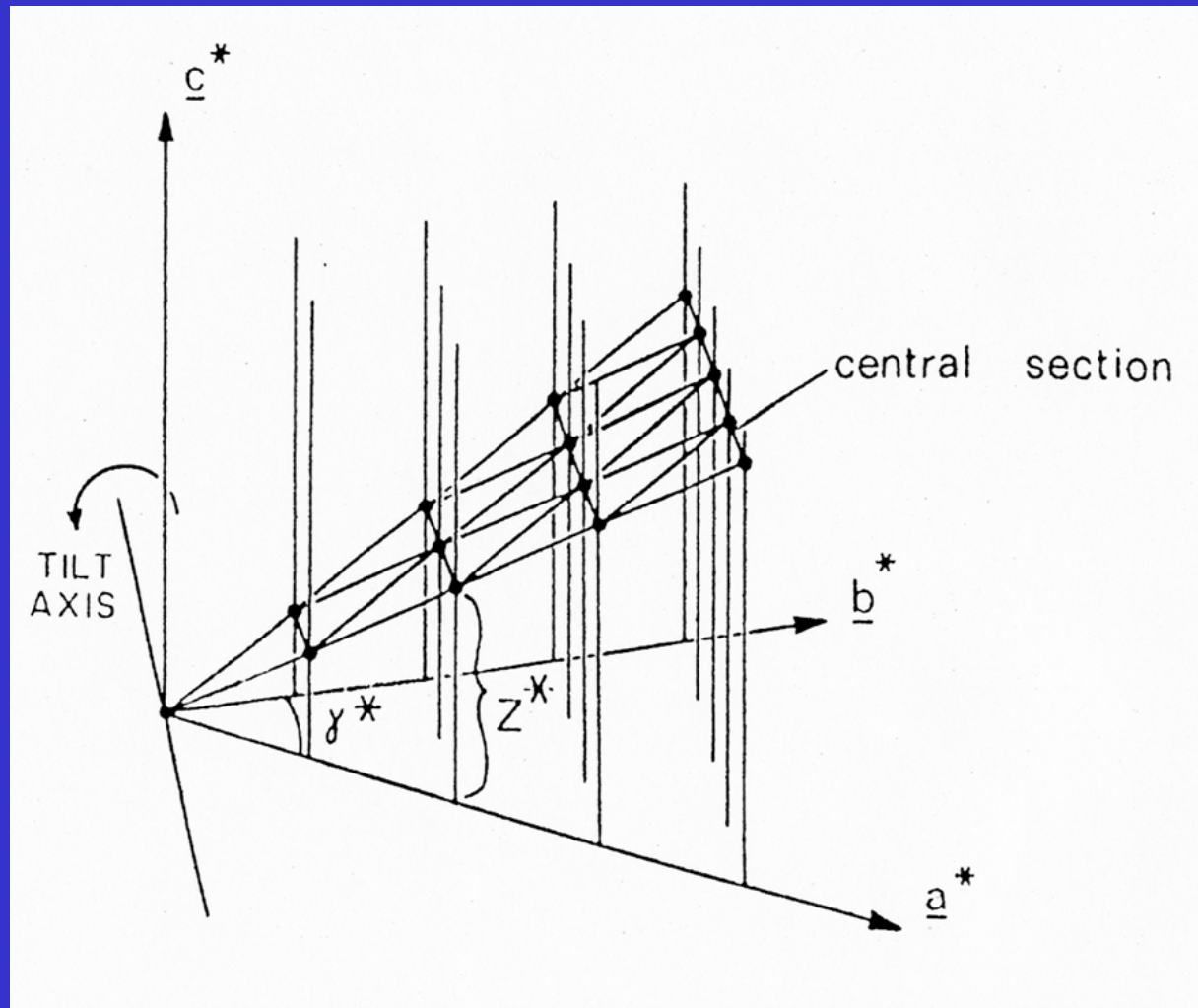
And so forth.....

Real space map obtained  
by Fourier summation

# Pictures of Tilted Crystals are Required for 3D-Structure Determination



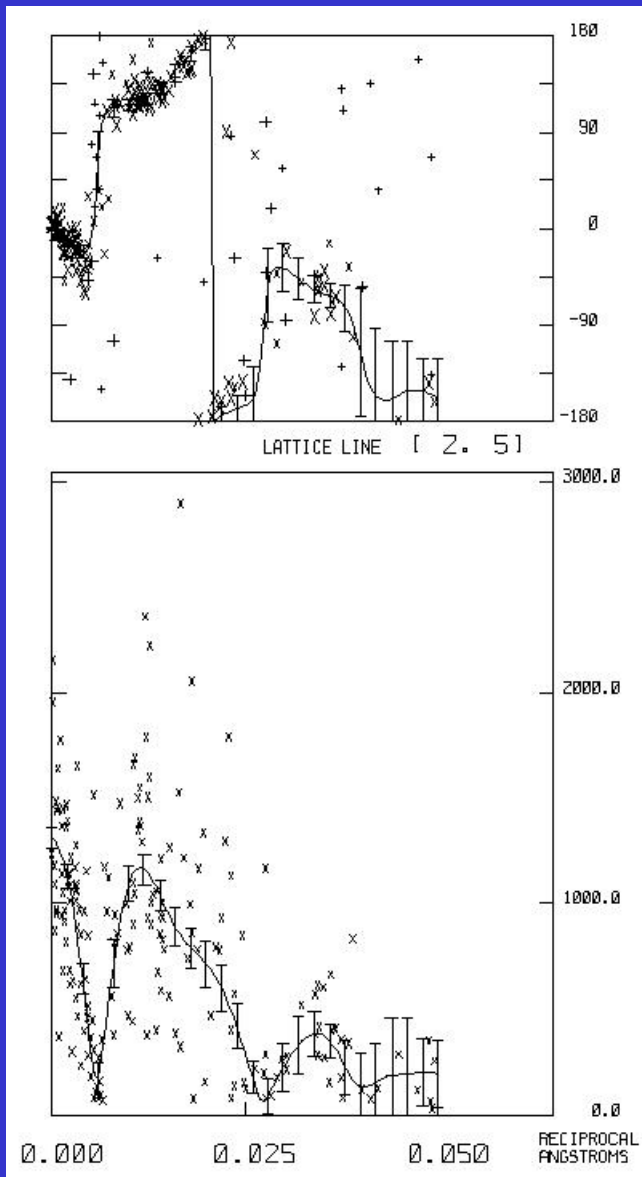
# Concept of Lattice Lines and Principle of Sampling their Data



Taken from:

Amos, Henderson and Unwin (1982), Prog. Biophys Molec Biol 39:183-231

# Example for a Lattice Line



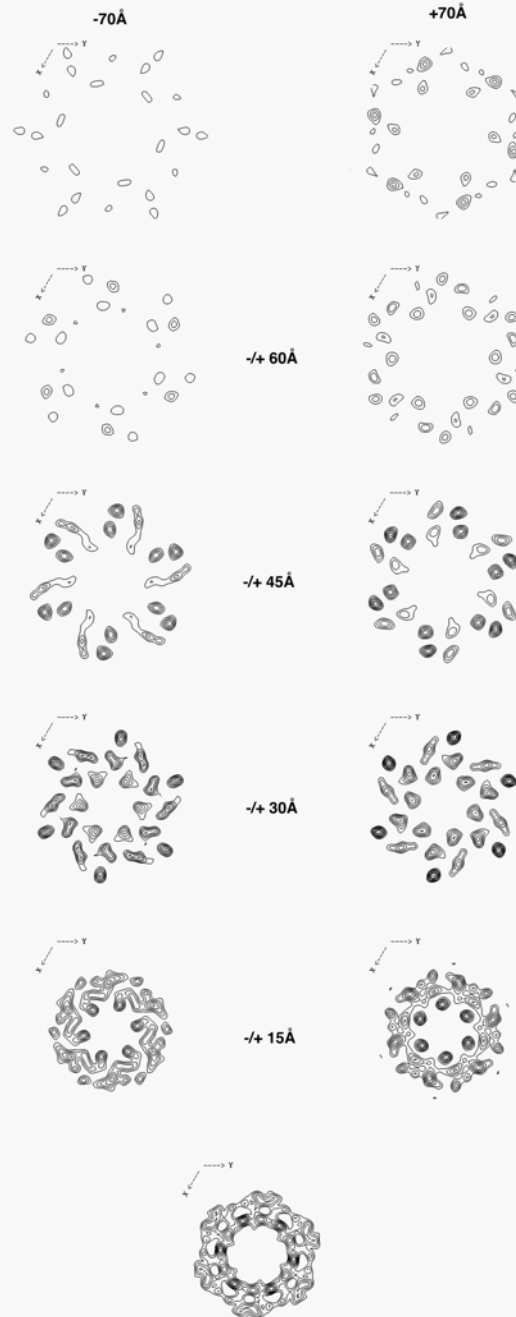
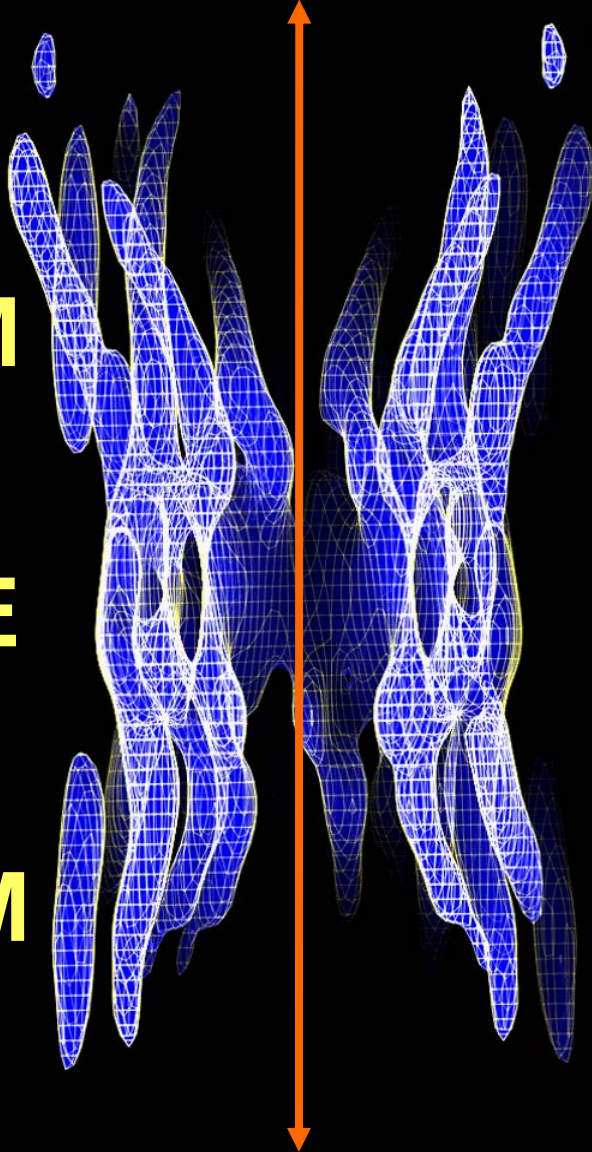
This figure shows the variation of phase (top panel) and amplitude (bottom) of the (2,5)-reflection of a gap-junction 2D-crystal as function of  $z^*$ .

The amplitudes were obtained from the calculated image transforms. In contrast to the phase information, **image derived amplitudes are very noisy** mostly because the image is modulated by the contrast transfer function of the objective lens (see page showing the calculated FT of an image). The effect of the CTF on amplitudes cannot be fully corrected, but, on the other hand **does not really matter that much because it is the phases that determine the structure.**

M

E

M



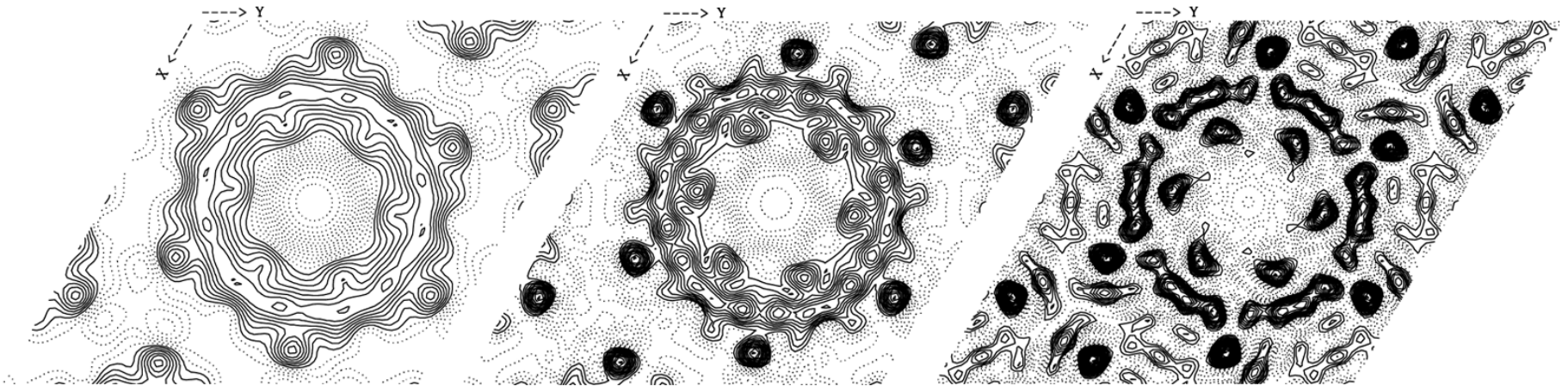
## 3D-Map of a Gap-Junction Intercellular Channel

Shown are a surface representation at  $\sim 7.5\text{\AA}$  resolution (left) and a montage of individual cross-sections (taken from a map at  $\sim 5.7\text{\AA}$  resolution). A total of  $\sim 42,000$  channel molecules were crystallographically averaged to obtain this structure.

**THE END**



## Effect of Image Sharpening (B-factor)



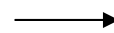
B=0

B = -350 Å<sup>-2</sup>

unity amplitudes

$$sf = e^{-B/4d^2}$$

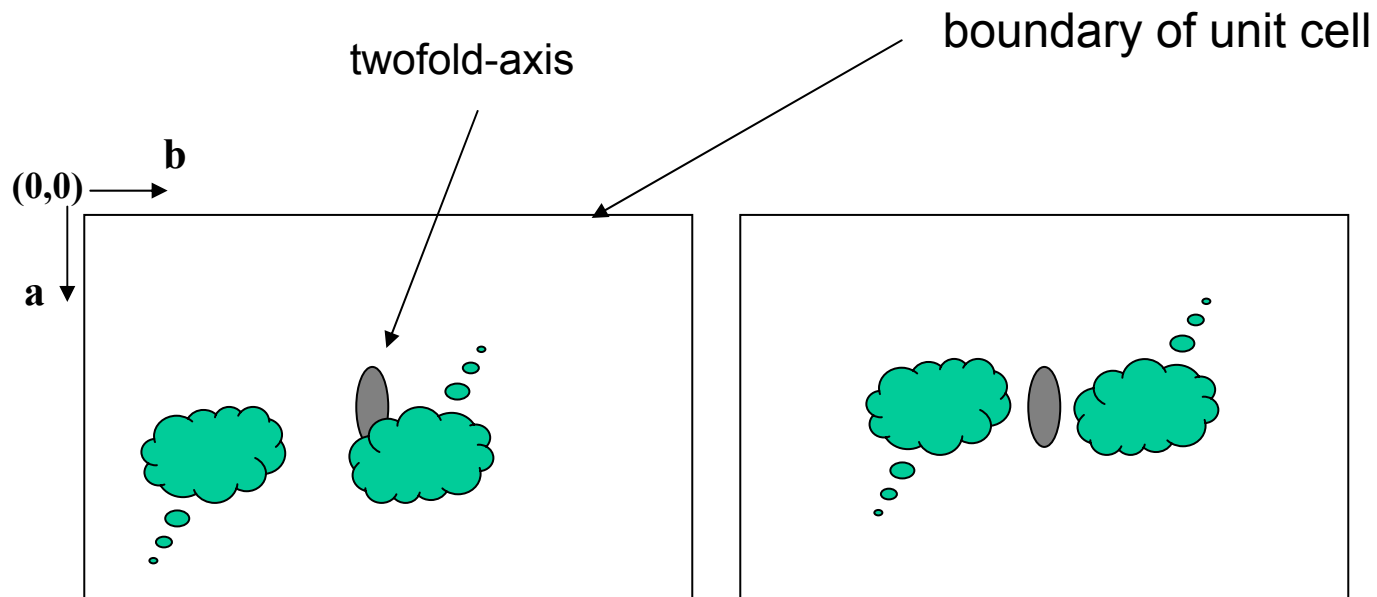
d = resolution [Å]



Example: 6Å, B=-350  
sf=11

Compare: unity amplitude  
sf=110 B=-680

# Principle of Symmetry Determination



Arbitrary location of dimer within unit cell. Apparent symmetry:  $p1$ ; no centrosymmetric phase constraints are obeyed!

Dimer centered about twofold axis of symmetry. Centrosymmetric phase constraints are obeyed = phases of all reflections are real (ie  $0^\circ/180^\circ$ )

**To test for 2-fold symmetry:** calculate phases for all reflections for all possible positions of the dimer within the boundary of the unit cell and check whether all phases become real. If this condition is met for any location within the unit cell then the crystal has at least twofold symmetry.

## Bringing Symmetry Related Reflections to a Common Phase Origin

<b>Reflection (h,k)</b>	<b>Phase extracted from calculated FT</b>	<b>Phase after shifting to proper phase origin</b>
(1,2)	198.0°	3.1°
(2,-3)	32.5°	1.0°
(-3,1)	198.2°	-0.4°

In p6 triplets of symmetry reflections all adopt the same phase when centered at the correct sixfold origin. Brought to this phase origin, the phase of each of the symmetry related reflections has to be close to either 0° or 180° because the inherent twofold symmetry (p6 symmetry implies p3 and p2 symmetry) constrains phases to be real in projection.

